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Theoretical Estimation of Power Generation Performance of Nano-Sheet Planar Lateral P-N Junction under Illumination

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Abstract – This paper proposes covalent-semiconductor-based lateral p-n junction film solar devices based on a theoretical model, and examines their power generation performance under illumination. The proposed theoretical model is implemented and tested in simulations. The results demonstrate that while Ge film devices have much lower performance at room temperature than Si film devices, this order is significantly reversed at temperatures below 250 K, which is very interesting. The obtained simulation results also reveal that the carrier generation characteristic of Ge film devices is very stable in terms of temperature variation in comparison to Si film devices. The simulation results suggest that thin-Si-film lateral p-n junction solar devices - implemented as multi-stacked solar devices formed on a transparent panel - are applicable to field sensor devices on the ground at temperatures lower than 300 K. However, thin-Ge-film lateral p-n junction solar devices are applicable to field sensor devices on satellites in space because the ambient temperature is lower than 250 K; again as a multi-stacked solar device formed on a transparent panel.

Keywords – Lateral p-n junction film solar device; Si; Ge; Sensor network; Low-energy applications.

1. INTRODUCTION

The fundamental of semiconductor pn junctions was proposed by Shockley [1] and many theoretical reconsiderations [2, 3] and device applications [4] have been advanced in the last 70 years. Recently, electrostatic physics for the planar sheet pn junction structure has been theoretically reviewed [5, 6], and a very interesting discussion was advanced. The theory mathematically describes the influence of geometry on the reduction in the depletion region across the pn junction. This theoretical advance seems very important from the viewpoint of future thin-film device applications of various pn junction structures.

The author proposed a pn-junction-based optical rotor that operates under external illumination [7-10]. Though the device was proposed as a mechanical part for Micro Electro Mechanical System (MEMS) applications [11-15], its full potentiality has yet to be examined in experiments. The MEMS applications envisaged [16-18] must have very low-power operation because they must work with very small batteries, not with ac power supplies. It is in this background that piezoelectric micro power generators are being investigated [19], where performance of ~0.01 μ J/cm³ has been demonstrated. Since the energy density is still small for most applications, further investigation is needed. In order to advance the performance of such power supplies, the author proposed an advanced Schenkel circuit for RF-ID

applications [20], but the idea remained to be confirmed as practical for low-energy and high-frequency circuits and systems.

Flexible electronics technologies are now being applied to solar battery devices formed on various films [21-23] because they have high potential for future mobile battery [24-26] and medical implantation applications [27]. Though past proposals of lateral p-n junction solar battery devices have already been compared [28], chalcogenides have become the focus of attnetion [29-36].

In this paper, the author theoretically examines the photovoltaic performance of the covalent-semiconductor-based planar lateral pn junction sheet to correct the paucity of interest in and research of it. In this study, not vertical, but the lateral pn junction structure is the focus of attention [7-10]. The vertical pn junction structure is very fatally weak against local threading defects. On the other hand, the lateral pn junction structure is very robust against such local defects. In addition, our study demonstrated that performance of a lateral thin-film Si pn junction device is almost insensitive to minority carrier lifetime [9]. Therefore, it is expected that such lateral pn junction devices are still promising for various applications.

Though the primary theoretical base was given in [10], some additional theoretical concepts are introduced and numerically evaluated. Here, various calculations are performed for Si and Ge from the viewpoint of applications, and the potential for device application is discussed.

2. THEORETICAL BASE OF SIMULATIONS

A physical image of the planar lateral p-n junction sheet is shown in Fig. 1, where the top view of the device is given in Fig. 1(a); it is assumed that the green light illumination is vertical to the sheet's surface. This configuration is quite different from those of conventional solar cells. A practical layout for applications is shown in Fig. 1(b). Device parameters assumed here are summarized in Table 1. The following mathematical analyses are based on Boltzmann's statistics.



Fig. 1. Nanometer-thick planar lateral p-n junction sheet on an insulator: a) basic configuration of the device and assumptions; b) image of practical application layout.

The generation and recombination process of holes, for example, in the p-type region under illumination, generally follows the continuity equation given by:

$$\frac{\partial p_p(x,t)}{\partial t} = G_p - \frac{p_p(x,t) - p_{p0}}{\tau_p} - p_p \mu_p \frac{\partial F}{\partial x} - \mu_p F \frac{\partial p_p}{\partial x} + D_p \frac{\partial^2 p_p}{\partial x^2}.$$
(1)

where p_{p0} , G_p , τ_p , F, μ_p , and D_p denote the hole concentration at thermal equilibrium, the generation rate of holes, the lifetime of holes, the local electric field, hole mobility, and the diffusion constant of holes, respectively. A similar equation is made for electrons in the n-type region.

The local electric field follows Poisson's equations as shown below:

$$-\frac{\partial F}{\partial x} = -\frac{q}{\varepsilon_s} (p_p - p_{po} - n_p + n_{p0}),$$

$$\cong -\frac{q}{\varepsilon_s} (p_p - N_A),$$
(2)

where *q* is the magnitude of elemental charge, ε_s is the permittivity of the semiconductor, N_A is the acceptor concentration in the p-type region, n_p is the electron concentration under illumination in the p-type region, and n_{p0} is the electron concentration at thermal equilibrium. As we assume the local gradient of $(p_p - N_A)$ is not so large, we have:

$$F \cong \frac{q}{\varepsilon_s} (p_p - N_A) L_{Dp}, \qquad (3)$$

where L_{Dp} is the Debye length of the p-type region.

At steady state, we have:

$$G_{p} - \frac{p_{p}(x,t) - p_{p0}}{\tau_{p}} - p_{p}\mu_{p}\frac{\partial F}{\partial x} - \mu_{p}F\frac{\partial p_{p}}{\partial x} + D_{p}\frac{\partial^{2}p_{p}}{\partial x^{2}} = 0.$$
(4)

Eq. (4) can be rewritten as:

$$D_{p}\frac{\partial^{2}p_{p}}{\partial x^{2}} - \left(\frac{qL_{Dp}\mu_{p}}{\varepsilon_{s}}\right) \left(2p_{p}\frac{\partial p_{p}}{\partial x} - N_{A}\frac{\partial p_{p}}{\partial x}\right) - \frac{p_{p}(x,t) - N_{A} - G_{p}\tau_{p}}{\tau_{p}} = 0.$$
(5)

The expression for
$$p_p(x)$$
 is replaced with the following excess hole concentration $P(x)$.

$$P(x) = p_p(x) - N_A - G_p \tau_p.$$
(6)

After some approximations are applied to Eq. (5)[10], we get the following solution:

$$2\frac{\sqrt{2C_{p}}}{A_{p}}\frac{P(0)}{P(0)+2\frac{\sqrt{2C_{p}}}{A_{p}}}\exp(\sqrt{2C_{p}}x)$$

$$P(x) = \frac{P(0)}{1-\frac{P(0)}{1-\frac{P(0)}{P(0)+2\frac{\sqrt{2C_{p}}}{A_{p}}}}\exp(\sqrt{2C_{p}}x)$$
(7)

$$P(0) = p_p(0) - N_A - G_p \tau_p, \tag{8}$$

$$A_p = \frac{2qL_{Dp}\mu_p}{D_p\varepsilon_s},\tag{9}$$

$$C_p = \frac{1}{D_p \tau_p}.$$
(10)

The theoretical expression for electrons in the n-type region is also written in similar algebra [10].

$$2\frac{\sqrt{2C_{n}}}{A_{n}}\frac{N(0)}{N(0)+2\frac{\sqrt{2C_{n}}}{A_{n}}}\exp(\sqrt{2C_{n}}x)$$

$$N(x) = \frac{1-\frac{N(0)}{1-\frac{N(0)}{N(0)+2\frac{\sqrt{2C_{n}}}{A_{n}}}}\exp(\sqrt{2C_{n}}x)$$
(11)

$$N(0) = n_n(0) - N_D - G_n \tau_n,$$
⁽¹²⁾

$$A_n = \frac{2qL_{D_n}\mu_n}{D_n\varepsilon_S},\tag{13}$$

$$C_n = \frac{1}{D_n \tau_n}.$$
(14)

where each parameter with suffix n' is the counterpart of that in the n-type region.

This study assumes the external illumination source power (P_{ph}) is given as:

$$P_{ph} = N_{ph} h \frac{c}{\lambda} \quad , \tag{15}$$

where *c* is the speed of light in a vacuum, λ is the wavelength, *h* is Plank's constant, and N_{ph} is the photon density. The carrier generation rate is given as:

$$G_n = G_p = \frac{N_{ph}}{t_s} (1 - R) l_c \left[1 - \exp\left(-\frac{t_s}{l_c}\right) \right], \qquad (16)$$

where *R* is the reflection fraction and l_c is the absorption length.

Since the external illumination yields excess electrons and holes in the film, they yield the following terminal-to-terminal potential difference (V_{out}).

$$V_{out} = \left(\frac{q}{\varepsilon_s}\right) \left[\int_{-L}^{0} \left(N_D + N(x)\right) x dx + \int_{0}^{L} \left(N_A + P(x)\right) x dx\right]$$
(17)

In addition, the internal resistance (R_{int}) under illumination is given as:

$$R_{\rm int} = \frac{L^2}{qWt_s \left[\mu_n \int_{-L}^0 (N_D + N(x)) dx + \mu_p \int_0^L (N_A + P(x)) dx \right]}$$
(18)

Using these calculation results, we can calculate, theoretically, the maximal voltage (V_{max}), the maximal load current (I_{max}), the efficiency, and fill factor [37], where temperature dependence of carrier mobility and band gap is given in the appendix.

3. CALCULATION RESULTS AND DISCUSSION

In this section, it is assumed that lateral sheet pn-junction devices are fabricated on a flexible film and glass. It is expected that those devices are applicable to wrist watches, IoT sensors, or satellites, making very low-energy operation systems essential. Therefore, it should be investigated whether the performance of the photovoltaic device satisfies the application requirements.

All simulations are performed using a home-made source code, not commercial software packages. Device parameters and physical parameters assumed here are summarized in Tables 1 and 2, respectively. The author's previous study demonstrated that the carrier generation and recombination process is insensitive to carrier lifetime values [9]. For simplicity, electrons and holes are assumed to have lifetimes of 10 ns.

Table 1. Parameters of the control device.					
Parameters	Values	Units			
Length of n-region and p-region (<i>L</i>)	1.0	μm			
Width of n-region and p-region (W)	10.0	μm			
Thickness (t_s)	100.0	nm			
Doping level of n-region (N_D)	1.0×10^{15}	/cm ³			
Doping level of p-region (N_A)	1.0×10^{15}	/cm ³			

Table 2. Physical parameters of the control device.					
Parameters	Values	Unit			
Lifetime of electrons (τ_{n0})	10.0	ns (@300K)			
Lifetime of holes (τ_{p0})	10.0	ns (@300K)			
Electron mobility (µn)					
Si	1400, 140	cm ² /V/s (@300K)			
Ge	3900, 390	cm ² /V/s (@300K)			
Hole mobility (μ_p)					
Si	500, 50	cm ² /V/s (@300K)			
Ge	1900, 190	cm ² /V/s (@300K)			
Coefficient of temperature dependence					
of mobility					
γ _n (Si)	2.42				
γ _p (Si)	2.20				
γ _n (Ge)	2.30				
γ _p (Ge)	1.60				
Illumination source power density (green light)	100.0	mW/cm ²			
l _c	0.55	μm			
R	0.5				
Band gap parameters					
α (Si)	4.73x10-4				
β (Si)	636.0	K			
α (Ge)	4.774x10-4				
β (Ge)	235.0	K			

Simulation results of the temperature dependence of the maximal voltage (V_{max} [V]), the maximal current density (I_{max} [mA/m²], the maximal power density (P_{max} [mW/m²]), efficiency, and fill factor for Si film solar panels is shown in Fig. 2; Fig. 2(a) is for the assumption of bulk quality parameters, while Fig. 2(b) is for the assumption of polycrystalline quality parameters. Comparing Fig. 2(a) with Fig. 2(b), we can find the following:

- a) V_{max} is insensitive to film quality, and $V_{max} \sim 0.1$ V (@300 K).
- b) I_{max} is sensitive to the film quality, and $I_{max} \sim 50 \ \mu\text{A}/\text{m}^2$ (@300 K) in Fig. 2(b).
- c) Except for V_{max} , most parameters are sensitive to carrier mobility.
- d) The efficiency is very low regardless of film quality.



Fig. 2. Temperature dependence of parameters for Si films: a) assumption of crystalline quality; b) assumption of polycrystalline quality.

Simulation results of the temperature dependence of the maximal voltage (V_{max} [V]), the maximal current density (I_{max} [mA/m²], the maximal power density (P_{max} [mW/m²]), efficiency, and fill factor for Ge film solar panels is shown in Fig. 3; Fig. 3(a) is for the assumption of bulk quality parameters, while Fig. 3(b) is for the assumption of polycrystalline quality parameters. Comparing Fig. 3(a) and Fig. 3(b), we can find the following:

a) V_{max} is insensitive to film quality, and $V_{max} \sim 1 \,\mu\text{V}$ (@300 K).

- b) I_{max} is sensitive to film quality, and $I_{max} \sim 0.1 \text{ mA/m}^2$ (@300 K) in Fig. 3(b).
- c) Except for V_{max} , most parameters are sensitive to carrier mobility.
- d) The efficiency is very low regardless of film quality.

A comparison of Figs. 2 and 3 reveals that Ge film devices have much lower performance at room temperature than the Si film devices, but much higher performance at temperatures below 250 K than the Si film devices, which is very interesting.



Fig. 3. Temperature dependence of parameters for Ge films: a) assumption of crystalline quality; b) assumption of polycrystalline quality.

Simulation results of the temperature dependence of the excess electron density (N(x) [/cm³]), the excess hole density (P(x) [/cm³]), the electron diffusion coefficient (D_n [cm²/s]), and the hole diffusion coefficient (D_p [cm²/s]) for Si film solar panels are shown in Fig. 4; Fig. 4(a) is for the assumption of bulk quality parameters, while Fig. 4(b) is for the assumption of polycrystalline quality parameters. Comparing Figs. 4(a) and 4(b), we can find the following:

a) Excess carrier densities, N(x) and P(x), are rapidly decreased at temperature higher than room temperature.

- b) Excess carrier densities, for the low film quality, are more rapidly decreased at temperature higher than room temperature than for the high film quality.
- c) Generation rates, G_n and G_p , are not sensitive to temperature.



Fig. 4. Temperature dependence of parameters for Si films at x = L/2: a) assumption of crystalline quality; b) assumption of polycrystalline quality.

Simulation results of the temperature dependence of the excess electron density $(N(x) \ [/\text{cm}^3])$, the excess hole density $(P(x) \ [/\text{cm}^3])$, the electron diffusion coefficient $(D_n \ [\text{cm}^2/\text{s}])$, the hole diffusion coefficient $(D_p \ [\text{cm}^2/\text{s}])$ for Ge film solar panels are shown in Fig. 5; Fig. 5(a) is for the assumption of bulk quality parameters, while Fig. 5(b) is for the assumption of polycrystalline quality parameters. Comparing Figs. 5(a) and 5(b), we can find the following:

- a) Excess carrier densities, N(x) and P(x), are basically insensitive to temperature independent of film quality unlike Si film devices.
- b) Generation rates, G_n and G_p , are not sensitive to temperature.

A comparison of Figs. 4 and 5 reveals that the carrier generation characteristic of the Ge film devices is basically stable against temperature variation unlike the Si film devices.



Fig. 5. Temperature dependence of parameters for Ge films at x = L/2: a) assumption of crystalline quality; b) assumption of polycrystalline quality.

Since various lateral thin-film pn-junction solar devices have already been investigated, their performances at room temperature are compared in Table 3, where no low-temperature performance data except this study is shown because such experiments and theoretical consideration are missing from past studies. It is known that low-temperature environments yield better performance than room temperature ones [38]. As shown in Table 3, at room temperature, Si-based lateral pn-junction solar devices have comparable performance to chalcogenide solar devices, except for efficiency. Its performance becomes better than that of chalcogenide solar devices except for efficiency at 150 K. On the other hand, the performance of Ge-based lateral pn-junction solar devices is inferior to that of chalcogenide solar devices. At 150 K, however, its performance becomes better than that of chalcogenide solar devices, except for efficiency at 150 K.

The above simulation results suggest that Si-based multi-stacked thin-film lateral pn junction solar devices formed on a transparent panel can be applied to field sensor devices on the ground at temperatures under 300 K. On the other hand, Ge-based thin-film lateral pn junction solar devices formed on a transparent panel can be applied to field sensor devices on satellites in space because the environmental temperature is lower than 250 K. In these

applications, it is expected that high-efficient peripheral circuits like Schenkel circuits [20, 39] and storage devices will also be required. Such lateral pn junction film devices [5, 6, 24] have already investigated for medical application [27] and others [40]. However, more investigation is needed from the viewpoint of availability of various semiconductor materials.

$\begin{array}{c c c c c c c c c c c c c c c c c c c $	Table 3. Photovoltaic metrices of lateral p-n junction solar devices (300K).							
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$	Material	Wave length [nm]	Vmax [V]	Fill factor	Efficiency	Ref.		
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$	WSe_2/MoS_2	(white light source)	0.22	0.39	0.2	[29]		
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$	WS_2/WSe_2	514	0.47			[33]		
$ \begin{array}{c c c c c c c c c c c c c c c c c c c $	MoSe ₂ /WSe ₂	532	0.9	0.2	0.08	[34]		
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$	WSe ₂	(white light source)	0.85	0.5	0.5	[35]		
$ \begin{array}{ c c c c c c c c } \hline bP & 640 & 0.05 & 0.1 & & [36] \\ \hline Si & 550 & 0.14 (300K) & 0.63 (300K) & 1.3x10^{-8} (300K) \\ \hline 0.77 (150K) & 0.92 (150K) & 1.2x10^{-7} (150K) \\ \hline Ge & 550 & \hline 7.9x10^{-7} (300K) & 0.99 (300K) & 2.5x10^{-13} (300K) \\ \hline This work & This work \\ \hline \end{array} $	WSe ₂	522	0.72			[31]		
Si 550 $0.14 (300 \text{K})$ $0.77 (150 \text{K})$ $0.63 (300 \text{K})$ $0.92 (150 \text{K})$ $1.3 \times 10^{-8} (300 \text{K})$ $1.2 \times 10^{-7} (150 \text{K})$ This workGe 550 $7.9 \times 10^{-7} (300 \text{K})$ $0.99 (300 \text{K})$ $2.5 \times 10^{-13} (300 \text{K})$ This workThis work	bP	640	0.05	0.1		[36]		
	Si	S: 550	0.14 (300K)	0.63 (300K)	1.3x10-8 (300K)	This work		
Ge 550 7.9x10-7 (300K) 0.99 (300K) 2.5x10-13 (300K) This work		550	0.77 (150K)	0.92 (150K)	1.2x10 ⁻⁷ (150K)	THIS WOLK		
Ge JJU IIIS WOIK	Ge	550	7.9x10 ⁻⁷ (300K)	0.99 (300K)	2.5x10 ⁻¹³ (300K)	This work		
0.29 (150K) 0.84 (150K) 1.2x10-7 (150K)		550	0.29 (150K)	0.84 (150K)	1.2x10 ⁻⁷ (150K)	THIS WOLK		

4. CONCLUSIONS

This paper proposed covalent-semiconductor-based lateral p-n junction film solar devices and examined their power generation performance under illumination. The comprehensive theoretical model previously proposed by the author was applied to the simulations conducted. The simulations assumed practical device parameters while the physical parameters were taken from a previous study.

Simulation results of the temperature dependence of various performance parameters were discussed for Si films and Ge films. For Si films, the following were revealed:

- a) V_{max} is insensitive to film quality, and $V_{max} \approx 0.1$ V (@300 K).
- b) I_{max} is sensitive to film quality, and $I_{max} \approx 50 \ \mu\text{A}/\text{m}^2$ (@300 K).
- c) Except for V_{max} , most parameters are sensitive to carrier mobility.
- d) The efficiency is very low regardless of film quality.

For Ge films, the following were revealed:

- e) V_{max} is insensitive to film quality, and $V_{max} \approx 1 \, \mu V$ (@300 K).
- f) I_{max} is sensitive to film quality, and $I_{max} \approx 0.1 \text{ mA}/\text{m}^2$ (@300 K).
- g) Except for *V_{max}*, most parameters are sensitive to carrier mobility.
- h) The efficiency is very low regardless of film quality.

The Ge film devices have much lower performance at room temperature than the Si equivalent, but that inferiority is reversed at temperatures below 250 K, which is very interesting. A comparison of Si films and Ge films revealed that the carrier generation characteristic of Ge film devices is basically stable against temperature unlike Si film devices.

These simulation results suggest that thin-Si-film lateral p-n junction solar devices in the multi-stacked arrangement formed on a transparent panel are applicable to sensor devices on the ground at temperatures of about 300 K. On the other hand, thin-Ge-film lateral p-n junction solar devices are superior as field sensor devices on satellites in space because the environmental temperature is lower than 250 K; again in the multi-stacked arrangement formed on transparent panels. Since lateral p-n junction film devices are already being

investigated for medical applications and others, it is clear that more investigation is needed from the viewpoint of the availability of various semiconductor materials for future applications.

APPENDIX: Temperature Dependence of Carrier Mobility, Band Gap, and Carrier Lifetime

Temperature dependence of carrier mobility is expressed as [41-43]:

$$\mu_n = \mu_{n0} \left(\frac{300}{T}\right)^{\gamma_n} \tag{A.1}$$

$$\mu_p = \mu_{p0} \left(\frac{300}{T}\right)^{\gamma_p} \tag{A.2}$$

Parameter values (μ_{n0} , μ_{p0} , γ_n , γ_p) are given in Table 2. In addition, temperature dependence of band gap is expressed as [41]:

$$E_G(T) = E_{G0}(0) - \frac{\alpha T^2}{T + \beta}$$
(A.3)

Parameter values (α , β) are also given in Table 2. Temperature dependence of lifetime is expressed as [41]:

$$\tau_n = \tau_{n0} \left(\frac{1}{e}\right) \exp\left(\frac{300}{T}\right) \tag{A.4}$$

$$\tau_p = \tau_{p0} \left(\frac{1}{e}\right) \exp\left(\frac{300}{T}\right) \tag{A.5}$$

Parameter values (τ_{n0} , τ_{p0}) are also given in Table 2.

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